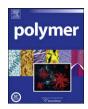
Polymer 50 (2009) 1626-1631

Contents lists available at ScienceDirect

Polymer



journal homepage: www.elsevier.com/locate/polymer



Polymer Communication

Crosslinking junctions of vulcanized natural rubber analyzed by solid-state NMR spectroscopy equipped with field-gradient-magic angle spinning probe

Seiichi Kawahara^{a,*}, Oraphin Chaikumpollert^a, Satoshi Sakurai^b, Yoshimasa Yamamoto^a, Keiichi Akabori^a

^a Department of Materials Science and Technology, Faculty of Engineering, Nagaoka University of Technology, Nagaoka, Niigata 940-2188, Japan ^b JEOL Ltd., 1-2-3 Musashino, Akishima, Tokyo 196-8556, Japan

A R T I C L E I N F O

Article history: Received 24 August 2008 Received in revised form 21 January 2009 Accepted 24 January 2009 Available online 7 February 2009

Keywords: NMR Natural rubber Crosslinking junctions

ABSTRACT

Crosslinking junctions of vulcanized natural rubber were analyzed by solid-state NMR spectroscopy equipped with a field-gradient high speed magic angle spinning probe. Resolution of ¹H and ¹³C NMR spectra and correlations between ¹H and ¹³C of the vulcanized natural rubber were investigated by oneand two-dimensional measurements, including inverse correlation measurements. The number of substitution of the carbon atom at the crosslinking junctions was determined by solid-state NMR spectroscopy.

Crown Copyright © 2009 Published by Elsevier Ltd. All rights reserved.

1. Introduction

Solid-state Nuclear Magnetic Resonance, NMR, spectroscopy is a powerful technique to analyze crosslinking junctions of elastomers and gels, since a chemical shift of signals appearing after crosslinking reflects a chemical environment of surroundings of the atoms of the crosslinking junctions [1,2]. The chemical environment may be positively analyzed by measuring correlations between ¹³C and ¹H. Thus, we must apply various pulse sequences to the analysis of the crosslinking junctions through NMR spectroscopy. However, in many literatures [3–8], only solid-state ¹³C NMR spectroscopy under high power ¹H-decoupling, i.e. dipolar decoupling, has been performed to analyze the crosslinking junctions, since a residual dipole-dipole interaction was so significant after crosslinking due to constrained motions of the elastomers and gels. To assign the signals appearing after vulcanization through solid-state ¹³C NMR spectroscopy, therefore, Mori and Köenig [5,6] adopted empirical correlations with shift factors to estimate values of chemical shifts of plausible crosslinking junctions proposed by Coran [9]. Small signals at 44, 50, 57 and 64 ppm of vulcanized natural rubber were assigned to tertiary carbons linking to S atoms [5,6], respectively.

In the previous works [10,11], we carefully assigned the signal at 44 ppm to secondary carbons and the signals at about 58 ppm to tertiary and guaternary carbons through latex-state NMR spectroscopy with distortionless enhancement by polarization transfer (DEPT) and attached proton test (APT) measurements. The assignment was proved by solution-state NMR spectroscopy for vulcanized liquid cis-1,4-polyisoprene and vulcanized squalene with DEPT, APT, 2-dimensional ¹H-¹³C correlation (HETCOR), 2-dimensional heteronuclear single coherence (HSQC) and 2-dimensional heteronuclear multiple bond correlation (HMBC) measurements. Furthermore, we proposed a method to prepare latex from vulcanized rubber sheets with a cryogenic sample crusher, in order to apply latex-state NMR spectroscopy to various vulcanized rubbers prepared in the solid state [11]. However, due to a complicated procedure to prepare the latex, it is anticipated to develop a novel solid-state NMR spectroscopy for the vulcanized rubber, in which various pulse sequences are applied to the analysis of the crosslinking junctions.

For the rubbery materials, high-resolution solid-state NMR spectroscopy without dipolar decoupling is performed with high speed magic angle spinning (HS-MAS) that is higher than about 20 kHz [12–18]. This may be attained even after vulcanization; hence, the residual dipole–dipole interaction of the vulcanized rubber is eliminated with HS-MAS. In this case, we may apply various pulse sequences, i.e. DEPT, APT, HETCOR and so forth to solid-state NMR spectroscopy for the vulcanized rubber.

^{*} Corresponding author. Tel.: +81 258 47 9301; fax: +81 258 47 9300. *E-mail address*: kawahara@mst.nagaokaut.ac.jp (S. Kawahara).

^{0032-3861/\$ –} see front matter Crown Copyright @ 2009 Published by Elsevier Ltd. All rights reserved. doi:10.1016/j.polymer.2009.01.062

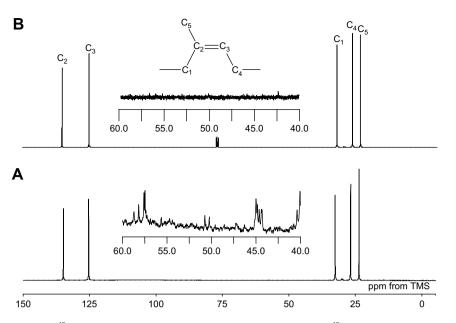


Fig. 1. Typical ¹³C NMR spectra: (A) solid-state ¹³C NMR spectrum for the vulcanized natural rubber, (B) solution-state ¹³C NMR spectrum for unvulcanized natural rubber. The solidstate ¹³C NMR measurement was performed with a 4 mm FG-MAS probe at 18 kHz in spinning rate. The solution-state ¹³C NMR measurement was performed with an NM-40TH5AT/ FG2SL probe at 12 Hz in spinning rate.

Furthermore, in order to detect very small signals of fewer amounts of crosslinking junctions, it is necessary to measure inverse correlations of ¹H with ¹³C, i.e. HSQC, HMQC and HMBC, which make possible to enhance sensitivity through solid-state NMR spectros-copy equipped with field-gradient HS-MAS (FG-HS-MAS) probe. In the present study, we apply newly developed FG-HS-MAS probe to the analysis of the crosslinking junctions of the vulcanized natural rubber through solid-state NMR spectroscopy, as the first trial. The crosslinking junctions of the vulcanized rubber were analyzed through solid-state NMR spectroscopy with DEPT, APT, HSQC and HMQC.

2. Experimental

Natural rubber latex used in this study was commercial high ammonia natural rubber latex of 60 w/w% dry rubber content (DRC). Vulcanization of natural rubber was made in latex stage. One example for the vulcanization of natural rubber was as follows; first, natural rubber latex (ca. 1000 ml) was prevulcanized with 88.5 g of 20 w/v% KOH solution, 17.5 g of 30%S/15%ZnO suspension and 99.5 g of 50 w/v% sodium di-n-butyldithiocarbamate solution at 40 °C for 4 h. Second, vulcanization of the prevulcanized natural rubber latex was carried out at 90 °C for 1-72 h. after dilution of the prevulcanized latex with water to adjust 20 w/w% DRC followed by adding 1 w/w% sodium dodecyl sulfate (SDS). The resulting vulcanized latex was filtered with 40 μ m mesh followed by centrifugation at 10,000g. Cream fraction of the latex was redispersed in 1 w/w% SDS solution to make 30 w/w% DRC latex and was washed by centrifugation, again. The washed cream fraction was redispersed in 0.1 w/w% SDS solution with deuterium oxide to make 50 w/w% DRC. A part of the cream fraction was coagulated and dried under reduced pressure. It was subjected to the solid-state NMR measurement.

Solid-state NMR spectroscopy was performed with a JEOL JNM-ECA600 FT-NMR spectrometer, operating at 600 and 150 MHz for ¹H and ¹³C, respectively. A spinning rate of the sample tube for the solid-state measurement was 18 kHz \pm 5 Hz. N₂ gas was used as MAS bearing and driving gas sources in order to protect samples

from oxidation and NMR probe head. The vulcanized natural rubber was loaded into a sample tube and its crosslinking junctions were analyzed with a 4 mm FG-MAS probe through solid-state NMR spectroscopy. Solution-state NMR spectroscopy was carried out with a IEOL INM-ECA400 FT-NMR spectrometer, operating at 399.7 and 100.4 MHz for ¹H and ¹³C, respectively. The vulcanized natural rubber latex with deuterium oxide was used for the measurements without further treatment with NM-40TH5AT/FG2SL probe. ¹H NMR, ¹³C NMR, DEPT and APT measurements were carried out at 323 K at pulse repetition times of 7 s and 5 s, respectively, since the liquid-like molecular motion should be accomplished at 323 K, which was about 100 K higher than the glass transition temperature of natural rubber [19]. The actual temperature of the surroundings in NMR probes was calibrated on the basis of the temperature dependence of the ²⁰⁷Pb chemical shift of Pb(NO₃)₂. Two-dimensional $^{1}H^{-13}C$ and heteronuclear multiple bond correlation measurements were made to collect two-dimensional hyper complex data. After weighting with shifted sine-bell function, the data was Fourier-transformed in the absolute value mode.

3. Results and discussion

Solid-state ¹³C NMR spectrum for the vulcanized natural rubber, measured with the FG-HS-MAS probe, is shown in Fig. 1, in conjunction with solution-state ¹³C NMR spectrum for unvulcanized natural rubber. Five major signals in the spectrum were

Table 1
Summary of relevant parameters for the spectra shown in Fig. 1.

	•	-
Figure	1A	1B
Sample	Vulcanized NR	Unvulcanized NR
Instruments (kHz)	150	100.4
Т (К)	323	323
No. of transients	90,000	90,000
Rec delay (s)	2.0	2.0
S/N	4882	5295
Half width (Hz)	17.05	11.47
Time (h)	76.5	76.5

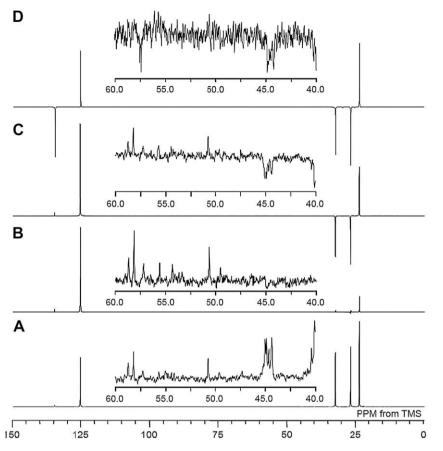


Fig. 2. Solid-state ¹³C NMR spectra with pulse sequences of distortionless enhancement by polarization transfer (DEPT) and attached proton test (APT) for the vulcanized natural rubber: (A) DEPT45, (B) DEPT90, (C) DEPT135, (D) APT.

assigned to C atoms of *cis*-1,4-isoprene units, according to the previous paper [11]. Values of half width of the signals and signal to noise (S/N) ratio of the solid-state ¹³C NMR spectrum for the vulcanized natural rubber were quite similar to those in the

solution-state ¹³C NMR spectrum for unvulcanized natural rubber, as shown in Table 1. The narrow half width and the sufficient S/N ratio of the solid-state ¹³C NMR spectrum for the vulcanized natural rubber may imply that high resolution was maintained for solid-

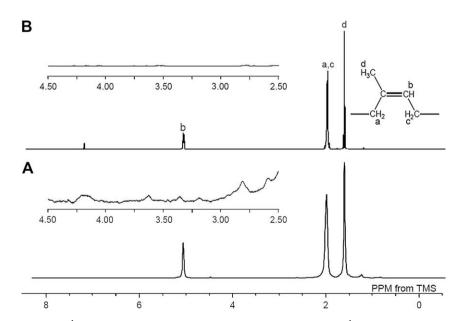


Fig. 3. Typical ¹H NMR spectra: (A) solid-state ¹H NMR spectrum for the vulcanized natural rubber, (B) solution-state ¹H NMR spectrum for the unvulcanized natural rubber. The solid-state ¹H NMR measurement was performed with a 4 mm FG-MAS probe at 18 kHz in spinning rate. The solution-state ¹H NMR measurement was performed with an NM-40TH5AT/FG2SL probe at 12 Hz in spinning rate.

 Table 2

 Summary of relevant parameters for the spectra shown in Fig. 3

Juminary of relevant parameters for the spectra shown in Fig. 5.			
Figure	3A	3B	
Sample	Vulcanized NR	Unvulcanized NR	
Instruments (kHz)	600	399.7	
T (K)	323	323	
No. of transients	128	128	
Rec delay (s)	5.0	4.2	
S/N	26,732	27,178	
Half width (Hz)	7.00	5.20	
Time (min)	13	13	

state ¹³C NMR spectroscopy even after vulcanization, since the dipole–dipole interaction between hetero-nuclei was eliminated by HS-MAS. This is distinguished from very low resolution of the solid-state NMR spectrum for the vulcanized natural rubber reported by Klüppel [17] and Köenig [3–8].

In Fig. 1, small signals appeared at 40, 44, 50 and 58 ppm in the solid-state ¹³C NMR spectrum for the vulcanized natural rubber, which were not shown in the solution-state ¹³C NMR spectrum for the unvulcanized natural rubber. To assign the signals, we applied DEPT measurement at 45° (DEPT45), 90° (DEPT90) and 135° (DEPT135) pulses and APT measurement to solid-state ¹³C NMR spectroscopy. Fig. 2 shows DEPT45, DEPT90 and DEPT135 spectra for the vulcanized natural rubber. The signals at 24, 26 and 32 ppm characteristic of methyl, methylene and methine carbons of *cis*-1,4-isoprene units were shown to be upward, upward and upward in the DEPT45 spectrum and they were almost null in the DEPT90 spectrum. On the other hand, in the DEPT135 spectrum, the signals were upward, downward and downward. Thus, the pulse width determined for DEPT measurements was confirmed to be correct to assign the small signals at 44, 50 and 58 ppm.

The small signals at 40 and 44 ppm were shown to be upward in the DEPT45 spectrum, null in the DEPT90 spectrum and downward in the DEPT135 spectrum; hence, they were assigned to secondary carbons. The signal at 50.5 ppm was assigned to quaternary carbon due to null signal in the spectra, while the signal at 50 ppm was assigned to tertiary carbon due to upward signals in the spectra, respectively. In contrast, the signal at 58 ppm was assigned to tertiary and quaternary carbons due to the very small upward signals in the DEPT45, DEPT90 and DEPT135 spectra and null signals. In Fig. 2, the APT spectrum is also shown for the vulcanized natural rubber. The APT spectrum represented downward signals at 40 and 44 ppm, upward and downward signals at 50 ppm, and upward and downward signals at 40 and 44 ppm to the secondary carbons and the signals at 50 and 58 ppm to the tertiary and quaternary carbons.

Fig. 3 shows solid-state ¹H NMR spectrum for the vulcanized natural rubber and solution-state ¹H NMR spectrum for the unvulcanized natural rubber with acquisition parameters listed in Table 2. Major signals at 1.7, 2.1 and 5.1 ppm in the spectra were assigned to methyl, methylene and unsaturated methine protons of cis-1,4-isoprene units, respectively. Values of half width and signal to noise (S/N) ratio of the signals in the solid-state ¹H NMR spectrum for the vulcanized natural rubber were a little bit larger and smaller, respectively, than the values of the half width and the S/N ratio of the signals in the solution-state ¹H NMR spectrum for the unvulcanized natural rubber. For instance, the value of half width of the signals in the solid-state spectrum was about 1.5 times as large as that in the solution-state spectrum. This may be explained to be due to a reduced effect of the dipole-dipole interaction between homo-nuclei in the solid-state ¹H NMR spectrum: that is, a major portion of the dipole-dipole interaction between homo-nuclei is eliminated by HS-MAS. The dipole-dipole interaction between homo-nuclei may be completely eliminated by extremely high speed MAS with smaller sample tube. However, we are not interested in the smaller sample tube and extremely high speed MAS, because the aim of the present study is to detect the correlation

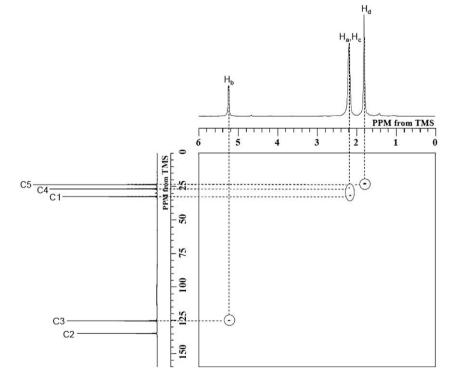


Fig. 4. HSQC spectra for the vulcanized natural rubber, obtained through solid-state NMR spectroscopy equipped with a 4 mm FG-MAS probe at 18 kHz in spinning rate for 1 h 2 min.

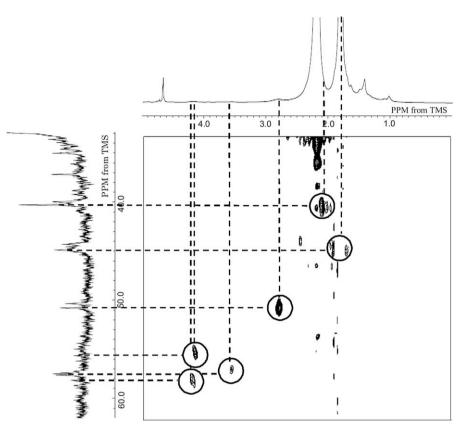


Fig. 5. HMQC spectra for the vulcanized natural rubber, obtained through solid-state NMR spectroscopy equipped with a 4 mm FG-MAS probe at 18 kHz in spinning rate for 71.5 h. HMQC measurement was performed in a selective region of chemical shift: 2–4 ppm in ¹H domain and 35–70 ppm in ¹³C domain.

between the small ¹³C and ¹H-signals assigned to the crosslinking junctions of vulcanized natural rubber. In Fig. 3, small signals at 3.4 and 4.2 ppm appeared in the solid-state ¹H NMR spectrum for the vulcanized natural rubber, but not in the solution-state ¹H NMR spectrum for the unvulcanized natural rubber. The signal at 3.4 ppm was assigned to aliphatic ¹H linking to -C-**CH**-S_x group and the signal at 4.2 ppm to unsaturated aliphatic ¹H linking to =C-**CH**-S_x group, according to the literature [20], which subscript x represents a number of S atoms.

Fig. 4 shows HSQC spectra obtained through solid-state NMR spectroscopy equipped with the FG-HS-MAS probe. The signals at 1.7, 2.1 and 5.1 ppm in the ¹H NMR spectrum were reasonably correlated with the signals in the ¹³C NMR spectrum; for instance, the signal at 1.7 ppm was correlated with the signal at 23 ppm, the signal at 2.1 ppm was with the signals at 26 and 32 ppm, and the signal at 5.1 ppm was with the signal at 135 ppm. However, no hetero-nuclei correlation between the small signals was detected in HSQC spectra due to unsatisfactory accumulation time for HSQC measurement.

To detect the hetero-nuclei correlation between the small signals, HMQC measurement was performed in a selective region of chemical shift: 2–4 ppm in ¹H domain and 35–70 ppm in ¹³C domain. Fig. 5 shows HMQC spectra obtained by the selective measurement. The ¹³C-signals at 40, 44, 50 and 58 ppm were well correlated to the ¹H-signals at 2.1, 1.7, 2.8 and more than 3.6 ppm. In the previous works [10,11], the signals at 40, 44 and 58 ppm of the vulcanized liquid *cis*-1,4-polyisoprene as a model were assigned through solution-state NMR spectroscopy with various pulse sequences, i.e. DEPT, APT, HETCOR, HSQC and HMBC. The signal at 40 ppm was assigned to C4 of *trans*-1,4-isoprene units, obtained by isomerization of *cis*-1,4-isoprene units. In contrast, the signals at

44 ppm in the ¹³C NMR spectrum were assigned to the secondary carbons adjacent to carbons linking to S atoms. The signals at 58 ppm in ¹³C NMR spectrum were assigned to the tertiary and quaternary carbons linking to S atoms. In the present work, the same results were obtained by solid-state NMR spectroscopy equipped with the FG-HS-MAS probe. Furthermore, the ¹³C-signals at 58 ppm were correlated to the ¹H-signals at 3.4 and 4.2 ppm, which were assigned to the $-C-CH-S_x$ group and $=C-CH-S_x$ group. More detailed assignments through solid-state NMR spectroscopy will be reported in our subsequent paper.

4. Conclusions

The crosslinking junctions of the vulcanized natural rubber were analyzed by solid-state NMR spectroscopy equipped with the FG-HS-MAS probe. The resolution of the solid-state ¹³C NMR spectra for the vulcanized natural rubber was sufficiently high, which was comparable to the resolution of the solution-state ¹³C NMR spectra for the unvulcanized natural rubber. The resolution of the solid-state ¹H NMR spectra for the vulcanized natural rubber was also similar to that of the solution-state NMR spectrum for the unvulcanized natural rubber. It was confirmed that the inverse correlation between ¹H and ¹³C-signals, assigned to the crosslinking junctions for the vulcanized natural rubber, was investigated by solid-state NMR spectroscopy equipped with the FG-HS-MAS probe. The signals at 40 and 44 ppm were assigned to C4 of trans-1,4-isoprene units and secondary carbons adjacent to carbons linking to S atom, respectively, and the signals at 58 ppm were to the tertiary and quaternary carbons linking to S atoms. Furthermore, the ¹³C-signals at 58 ppm were distinguishably

correlated to the ¹H-signals at 3.4 and 4.2 ppm, which were assigned to the -C-**CH**-S_x group and =C-**CH**-S_x group, respectively.

Acknowledgments

This work was supported in part by a Grant-in-Aid (12416) for the Development of Innovation and the 21st Century COE Program for Scientific Research from The Ministry of Education, Culture, Sports, Science and Technology, Japan.

References

- [1] Bovey FA. NMR of polymers. Academic Press; 1996.
- [2] Ando I, Asakura T. Solid state NMR of polymers. Elsevier Science; 1998.
- [3] Zaper AM, Köenig JL. Rubber Chem Technol 1987;60:252–77.
- [4] Mori M, Köenig JL. Rubber Chem Technol 1995;68:551–62.
- [5] Mori M, Köenig JL. Rubber Chem Technol 1997;70:671–80.
- [6] Mori M. Rubber Chem Technol 2003;76:1259–75.

- [7] Mori M, Köenigh JL. J Appl Polym Sci 1998;70:1391-9.
- [8] Patterson DJ, Köenigh JL, Shelton JR. Rubber Chem Technol 1983;56: 971-94.
- [9] Coran AY. In: Mark JE, Erman B, Eirich FR, editors. Science and technology of rubber. 2nd ed. San Diego: Academic Press; 1994 [chapter 7].
- [10] Ukawa J, Kawahara S, Sakai J. J Polym Sci Part B Polym Phys 2007;45:1003-9. [11] Kawahara S, Jinta U, Sakai J, Yamamoto Y, Isono Y. Rubber Chem Technol
- 2007;80:751–61. [12] Wang M, Bertmer M, Demco DE, Blumich B. | Phys Chem 2004;B108:10911–8.
- [13] Cohen-Addad JP, Vogin R. Phys Rev Lett 1974;33:940–3.
- [14] Saalwachter K, Herrero B, Lopez-Manchado MA. Macromolecules 2005; 38:9650-60.
- [15] Mawell RS, Chinn SC, Solyom D, Cohenour R. Macromolecules 2005;38: 7026-32.
- [16] Menge H, Hotopf S, Heuert U, Schneider H. Polymer 2000;41:3019–27.
- [17] Klüppel M, Menge H, Schmidt H, Schneider H, Shuster RH. Macromolecules 2001;34:8107–16.
- [18] Simon G, Baumann K, Gronski W. Macromolecules 1992;25:3624-8.
- [19] Shaw MT. In: Shaw MT, MacKnight WJ, editors. Introduction to polymer viscoelasticity. 3rd ed. John Wiley & Sons; 2005.
- [20] Frust A, Pretsch E. Anal Chem Acta 1990;229:17-25.